

ly. This is, however, only an apparent cross section. The true capture cross section σ_p' is larger and is related to σ_p by

$$\sigma_p' = \sigma_p \exp(\phi_B/kT), \quad (1)$$

because the carriers have to overcome the grain-boundary potential ϕ_B to be captured. From these measurements it was also found that for saturating pulse widths the concentration of trap H_1 was $\sim 10^{14} \text{ cm}^{-3}$.

DLTS signals of hole traps in comparable material to Fig. 1(a) obtained with mesa n^+p structures looked very similar to those of Fig. 1, indicating the presence of the same traps. Peak H_4 , however, was less prominent, probably a result of the cell fabrication process. In addition to hole traps, electron traps were also observed with the n^+p mesa structures. Figure 3(a) is their DLTS signal, and Fig. 3(b) their Arrhenius plots. Four electron traps could be identified at levels $E_1: E_c - 0.44 \text{ eV}$, $E_2: E_c - 0.62 \text{ eV}$, $E_3: E_c - 0.32 \text{ eV}$, and $E_4: E_c - 0.16 \text{ eV}$, where E_c is the bottom of the conduction band. The capture cross sections of these levels were calculated from their Arrhenius plots and are $\sim 10^{-21}$, $\sim 10^{-15}$, $\sim 10^{-19}$, and $\sim 10^{-23} \text{ cm}^2$, respectively.

It was previously stated that the spectrum shown in Fig. 1(a) was obtained from material containing active grain boundaries. Indeed, no other regions showed the presence of hole traps, implying that all the hole (i.e., majority) traps found to exist in this material are grain-boundary related. Electron traps at large concentrations were found to exist near the bottom of the ingot and in regions close to the crucible. DLTS signals of minority traps were also obtained from regions well away from the crucible and the bottom of the

ingot, and it was found that the minority trap density in such regions is much lower. This is in agreement with results of diffusion-length measurements performed at Solarex Corp. where it was found that the minority carrier diffusion length was reduced in the lower regions of the ingot.⁸

The main conclusion of this work is that in p -type cast polycrystalline silicon, the majority-carrier traps are mostly grain-boundary related, whereas the minority-carrier traps are found in all regions of the ingot. The minority-carrier trap concentration is larger in areas near the crucible and at the bottom of the ingot, and appears to be growth related.

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¹A number of papers dealing with this subject are found in *Electronic and Optical Properties of Polycrystalline or Impure Semiconductors and Novel Silicon Growth Methods*, edited by K. Y. Ravi and B. O'Mana (Proceedings of Electrochemical Society, St. Louis, Missouri, 1980), Vol. 80-5.

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Characterization of free jet expansion of SF₆ molecules

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In this work we present experimental measurements of the mean velocity of SF₆ molecules in a pulsed-free jet as a function of the distance from the expansion orifice. A pyroelectric detector was used to measure the time of flight of the molecules in the jet. A comparison of the experimental results with theoretical values for the mean velocity taking the temperature dependence of the specific-heat ratio into account leads to the conclusion that the freezing of rotational, vibrational, and translational temperature is reached at considerable distances from the expansion orifice.

Molecular beams have been used in many branches of scientific research. In fundamental physics they are used to measure magnetic, electric, and spin momentum in atoms and molecules.¹ In chemistry, gas-kinetics and gas-dynamics, they were used to study reaction kinetics.¹⁻⁶ In the beginning of this decade, molecular beams were used in photo

chemical research to study the interaction of molecules and clusters with intense electromagnetic radiation fields from coherent sources.⁷⁻¹²

There are several conventional methods to produce a molecular beam¹: (a) effusive ovens which produce low-energy and low-density beams; (b) ionic sources with charge

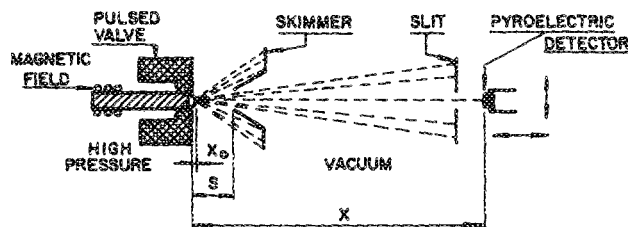


FIG. 1. Schematic of experimental setup.

exchange for highly energetic and low density beams; (c) shock tubes which produce pulsed beams of high energy and density; and (d) the gas-dynamic methods for production of beams of medium densities and energies.

Due to the ease of construction, wide range, and controllability of energies and densities, in addition to low translational, vibrational, and rotational temperatures obtained, the gas-dynamic methods have been largely used to study SF_6 molecules in free molecular jets.⁷⁻¹⁰

In this work, a molecular beam of SF_6 molecules was produced by a supersonic expansion. The gas was pressurized to about 10 atm in a stagnation chamber made out of stainless steel and large enough as to assure constant pressure during valve operation. A pulsed valve, as indicated in Fig. 1, with a 1-mm-diam circular orifice, connects the high-pressure section to a high vacuum chamber also made out of stainless steel allowing the gas to expand during an opening time of about 80 μs (FWHM). A high vacuum turbomolecular pumping system with 2000 l/s capacity was used to keep the pressure in the vacuum chamber below 10^{-5} mbar so that the background has no influence on the SF_6 beam. Skimmer and slit are placed inside the high-vacuum section to get well collimated molecular beams. A pyroelectric detector made out of PVF_2 organic film (6 μm thick, area of 7.0 mm^2) is placed in front of the molecular beam (Fig. 1) in a sliding mount which allows to change the distance x in Fig. 1. The energy of the incident molecular jet is converted into heat at the surface of the detector¹³ and an electric signal is

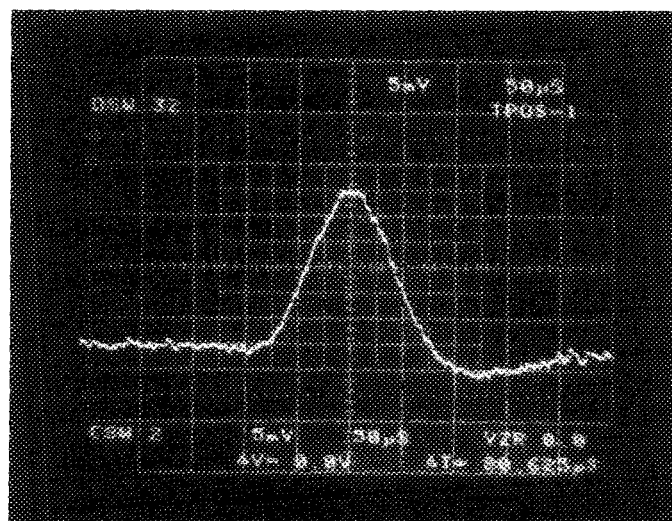


FIG. 2. Typical pulse from pyroelectric detector that shows the opening time of a pulsed valve.

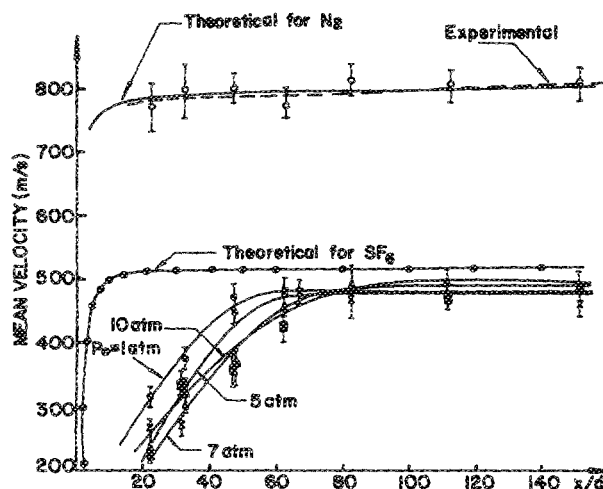


FIG. 3. The theoretical and experimental results of mean velocity vs distance from the orifice in the center jet.

therefore generated.¹⁴ A typical pulse obtained in a screen of an oscilloscope is shown in Fig. 2. As the detector-amplifier system has a time constant of approximately 10 μs , the pulse width in Fig. 2 is determined by a combination of the valve opening time distribution and the spatial spread of the molecules in the jet. The mean velocity is obtained by measuring the time of flight of the molecules between the orifice and the detector. The position of the detector was changed from a distance of 11.4 to 152.0 mm to the expansion orifice in steps of about 10 mm. The experimental results for the mean velocity as a function of the position are shown in Fig. 3 for different pressures in the stagnation chamber. An experiment with N_2 was realized in order to calibrate TOF measurements. The results are shown in Fig. 3, where the very good agreement with the theory validates the experimental apparatus. The theoretical curves were obtained as follows:

The translational temperature and density of the molecules in the jet are obtained from the relations⁴:

$$T = T_0 [1 + (\gamma - 1) M^2 / 2]^{-1} \quad (1)$$

and

$$n = n_0 [1 + (\gamma - 1) M^2 / 2]^{1/(\gamma - 1)}, \quad (2)$$

TABLE I. Dependence of the specific heat ratio on the temperature for SF_6 molecules.

T (K)	γ	T (K)	γ
20.0	1.333 334	170.0	1.164 072
30.0	1.333 328	180.0	1.153 939
40.0	1.333 130	190.0	1.144 994
50.0	1.331 749	200.0	1.137 097
60.0	1.327 488	210.0	1.130 119
70.0	1.319 118	220.0	1.123 939
80.0	1.306 574	230.0	1.118 454
90.0	1.290 740	240.0	1.113 572
100.0	1.272 915	250.0	1.109 214
110.0	1.254 367	260.0	1.105 312
120.0	1.236 107	270.0	1.101 808
130.0	1.218 813	280.0	1.098 651
140.0	1.202 869	290.0	1.094 477
150.0	1.188 438	300.0	1.093 217
160.0	1.175 531		

where M is the Mach number, γ is the specific-heat ratio, and a zero subscript refers to the stagnation conditions. Here γ and M are functions of the temperature (T) and consequently of the distance x from the expansion orifice. Using the method of characteristics,¹⁵ a relation between M , x , and γ is written as^{4,16}

$$M = A \left(\frac{x - x_0}{D} \right)^{\gamma-1} - \frac{1}{2} \frac{(\gamma+1)/(\gamma-1)}{A [(x - x_0)/D]^{\gamma-1}}, \quad (3)$$

with⁴

$$x_0/D = 112.5390 \exp(-3.9196\gamma) \quad (4)$$

and

$$A = 202.2092 - 317.7597\gamma + 21.1011\gamma^2 + 172.2215\gamma^3 - 69.7483\gamma^4 \quad (5)$$

for $1.35 \leq \gamma \leq 1.05$.

The temperature dependence of γ is obtained from the semi-classical partition function (Z) which allows us to calculate $\gamma(T)$. We used

$$C_p = - \frac{\partial}{\partial T} \frac{\partial \ln Z}{\partial (1/KT)} \quad (6)$$

to generate the data of Table I for SF_6 molecules. Using the equations above and the fact that $M = 1$ at $x = 0$ at the orifice and for fixed stagnation conditions, T and n can be calculated for increasing values of the Mach number. Using an iterative process, we calculate the position x at which a set of compatible variables T , γ , M , and n are given. From this, the theoretical curve of Fig. 3 was generated. The discrepancy between theoretical and experimental curves for the mean velocity in the region $x/d \leq 60$ is attributed to the fact that thermal equilibrium is not maintained in this region and so that a discharge coefficient must be considered. For $x/d > 60$ region, theoretical values are greater than experimental ones because some energy is still stored in SF_6 molecules as vibra-

tional energy and therefore maximum theoretical translational energy is not reached. This is due to the large number of degrees of freedom of SF_6 molecules. Therefore, near the orifice, relations (1)–(5) are valid only for light molecules or atoms.⁴ The SF_6 pressure dependence of the experimental results is explained by the fact that the collision region increases with the pressure of the stagnation chamber resulting as a consequence of a broader region where thermal equilibrium is not maintained ($x/d < 60$). On the other hand, this pressure effect causes a higher rate of vibrational to translational energy transfer so that in the region $x/d > 60$ higher molecular speeds are reached corresponding to higher stagnation pressures.

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Photoelectronic properties of HgI_2

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Two types of photoelectronic measurements have been used to determine the properties of HgI_2 crystals and photodetector cells: thermally stimulated conductivity, and spectral response of photoconductivity in both the lateral and transverse geometries. Characteristic trap depths are identified that compare well with previously reported values, and a decrease in apparent density with time and temperature cycling is described. Analysis of photoconductivity spectral response using the DeVore model for lateral contacts and the Goodman model for transverse contacts allows the determination of photoelectronic parameters for the material.

Mercuric iodide is a material of interest for applications to solid-state photodetection of gamma radiation in combination with a scintillator such as calcium tungstate or thallium-activated sodium iodide. A recent symposium¹ has summarized much of the work in recent years. Other investi-

gations have been carried out on the photoelectronic properties,^{2,3} transport properties,⁴ and photomagnetoelectric effect^{5,6} of HgI_2 . One of the problems with HgI_2 detectors is an unexplained variation in sensitivity of the HgI_2 detector with time.⁷